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NOVEL POLYIMIDE AND CIRCUIT SUBSTRATE COMPRISING THE SAME

# Abstract:

The invention is a polyimide, random copolymer having repeating units of the formula (1): <CHEM> wherein R1 and R2 each represent a divalent group selected from <CHEM> and R1 and R2 may be the same or different; and x=0.60 to 0.80, y+z=0.40 to 0.20, and x+y+z=1.00, and a linear expansion coefficient at 100 to 200 DEG C is in the range of from 10 to 20 ppm/K. This is a useful polyimide that can be a polyimide circuit substrate material capable of keeping flat, neither shrinking nor expanding in its laminate.

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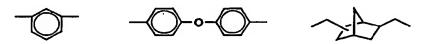
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# (54) NOVEL POLYIMIDE AND CIRCUIT SUBSTRATE COMPRISING THE SAME

(57) The invention is a polyimide, random copolymer having repeating units of the formula (1):

wherein R1 and R2 each represent a divalent group selected from



and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00, and a linear expansion coefficient at 100 to 200°C is in the range of from 10 to 20 ppm/K. This is a useful polyimide that can be a polyimide circuit substrate material capable of keeping flat, neither shrinking nor expanding in its laminate.

#### Description

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#### **TECHNICAL FIELD**

[0001] The present invention relates to a polyimide that comprises specific repeating structural units and has good thermal resistance, mechanical properties and chemical resistance. Additionally, for polyimide/metal leaf laminates that are said to be especially important in the field of electronic materials, the invention relates to a useful polyimide which is so designed that its linear expansion coefficient approximates that of metal leaf and which can be a polyimide circuit substrate material capable of keeping flat, neither shrinking nor expanding in its laminate.

#### **BACKGROUND ART**

[0002] The linear expansion coefficient (thermal expansion coefficient) of most organic polymers is at least 50 ppm/ K even in a temperature range not higher than the glass transition temperature thereof, and is much higher than that of metals and inorganic substances. Therefore, metal/polymer composites, for example, metal leaf/organic polymer laminates involve serious problems of warping, deformation, delamination, cracking of the polymer layer and disruption of the substrate itself because of the difference in the linear expansion coefficient between the constituent components.

[0003] In that problematic situation, polyimide having good thermal resistance and good mechanical properties has heretofore been used as an organic polymer that solves the problems.

[0004] For example, the polyimide represented by the formula (A):

has been known as a typical example. This is a resin having good thermal resistance and good mechanical properties, and is widely used for electronic materials. However, its linear expansion coefficient is about 40 ppm/K (in the temperature range of 100 to 200°C), and the polyimide is still unsatisfactory for a constituent component of laminates.

[0005] On the other hand, the polyimide represented by the formula (B):

$$\begin{array}{c|c}
 & \circ & \circ & \circ \\
 & \circ & \circ &$$

also has good thermal resistance and good mechanical properties, and its linear expansion coefficient is extremely low. Concretely, however, the linear expansion coefficient of the polyimide is at most 10 ppm/K (in the temperature range of 100 to 200°C), and contrary to the above, it is lower than the linear expansion coefficient of metal.

[0006] Recently, other polyimides having a low linear expansion coefficient (low stress) have been developed to solve these problems. For example, in Japanese Patent Laid-Open Nos. 191830/1988, 199236/1988, 16829/1989, 20232/1989, 33134/1989 and 38437/1989, alkyl chains such as methyl groups are introduced into rigid polyimide main chain skeletons to lower the linear expansion coefficient of the thus-modified polyimides. However, these polyimides (or their precursors, polyamic acids) are problematic in that the polymer component deposits in the varnish thereof when stored long. Another problem with them is that, since they have many alkyl groups introduced thereinto, the methyl groups therein are oxidized when they are exposed to high temperatures.

[0007] In Japanese Patent Laid-Open Nos. 153934/1990 and 251584/1990, disclosed are polyimides having rigid hetero rings such as imidazole introduced thereinto. However, the hetero ring-having monomers for them require complicated steps in their production, and their costs inevitably increase.

[0008] A polyimide having a controlled linear expansion coefficient useful for polyimide/metal leaf has not as yet been realized. In that situation, a technique for controlling the linear expansion coefficient of polyimide through stretching and orientation is known. However, the technique requires an additional secondary step, and is therefore problem-

atic in practical use.

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[0009] As mentioned hereinabove, a polyimide satisfactory for laminates of polyimide/metal leaf is not as yet found out, and a polyimide that solves the problems noted above is desired.

#### DISCLOSURE OF THE INVENTION

[0010] The object of the present invention is to provide a polyimide having a suitable linear expansion coefficient that corresponds to the linear expansion coefficient of metal leaf (in this case, in the range of 10 to 20 ppm/K at the temperature range of 100 to 200°C) in addition to good properties inherent to polyimides, that is, good thermal resistance, mechanical properties and chemical resistance. More precisely, it is to provide a useful polyimide which can be a polyimide circuit substrate material capable of keeping flat, neither shrinking nor expanding in its laminate.

[0011] The present inventors have assiduously studied to attain the object mentioned above and, as a result, have found that a polyimide obtained through thermal imidation of a polyamic acid copolymer, which is prepared by reacting a) pyromellitic acid dianhydride selected for a tetracarboxylic acid dianhydride component with b) a diamine component of essentially paraphenylenediamine combined with 4,4'-oxydianiline or two diamines selected from 4,4'-oxydianiline, metaphenylenediamine and diaminomethyl-bicyclo[2.2.1]heptanes in a specific composition ratio, satisfies the necessary linear expansion coefficient and therefore solves the problems noted above. On the basis of this finding, the present invention has been completed.

[0012] Specifically, the invention relates to the following:

1) A polyimide, which is a random copolymer having repeating units represented by the formula (1):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C;

2) The polyimide of above 1, which is a random copolymer having repeating units represented by the formula (2):

wherein x1 = 0.60 to 0.80, y1 = 0.40 to 0.20, and x1 + y1 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C;

3) The polyimide of above 1, which is a random copolymer having repeating units represented by the formula (3):

$$\left( \bigcirc \stackrel{\circ}{\mathsf{N}} \stackrel{\circ}{\mathsf{S}} \stackrel{\circ}{\mathsf{N}} \right)_{\mathsf{X}2} \left( \bigcirc \stackrel{\circ}{\mathsf{N}} \stackrel{\circ}{\mathsf{S}} \stackrel{\circ}{\mathsf{N}} \right)_{\mathsf{X}2} \left( \bigcirc \stackrel{\circ}{\mathsf{N}} \right)_{\mathsf{X}2} \left( \bigcirc \stackrel{\circ}{\mathsf{N}} \stackrel{\circ}{\mathsf{N}} \right)_{\mathsf{X}2}$$

wherein x2 = 0.60 to 0.80, y2 = 0.35 to 0.05, z2 = 0.05 to 0.15, and x2 + y2 + z2 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C;

4) The polyimide of above 1, which is a random copolymer having repeating units represented by the formula (4):

$$\left( \bigcirc \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{X3} \left( \stackrel{\circ}{N} \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{Y3} \left( \bigcirc - \bigcirc - \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{Z3} (4)$$

wherein x3 = 0.65 to 0.75, y3 = 0.30 to 0.10, z3 = 0.05 to 0.15, and x3 + y3 + z3 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C;

5) The polyimide of above 1, which is a random copolymer having repeating units represented by the formula (5):

wherein x4 = 0.65 to 0.75, y4 = 0.05 to 0.15, z4 = 0.30 to 0.10, and x4 + y4 + z4 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C;

6) The polyimide of above 1, of which the precursor, polyamic acid has a number-average molecular weight (Mn) of at least 40,000, a weight-average molecular weight (Mw) of at least 60,000, and a molecular weight distribution (Mw/Mn) being in the range of 1.6 to 2.3;

7) A process for preparing the polyimide of above 1, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of a diamine of the formula (7):

and from 40 to 20 mol% of diamines of the formulae (16) and (17):

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$$H_2N - R_1 - NH_2 \tag{16}$$

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

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and  $R_1$  and  $R_2$  may be the same or different, in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units represented by the formula (15):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00; 8) A process for preparing the polyimide of above 2, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of the formula (7):

and from 40 to 20 mol% of 4,4'-oxydianiline of the formula (8):

$$H_2N \longrightarrow O \longrightarrow NH_2$$
 (8)

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units represented by the formula (9):

wherein x1 = 0.60 to 0.80, y1 = 0.40 to 0.20, and x1 + y1 = 1.00;

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9) A process for preparing the polyimide of above 3, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of the formula (7):

$$H_2N \longrightarrow NH_2$$
 (7)

from 5 to 15 mol% of 4,4'-oxydianiline of the formula (8):

$$H_2N - \bigcirc -O - \bigcirc -NH_2$$
 (8)

and from 35 to 5 mol% of metaphenylenediamine of the formula (10):

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units represented by the formula (11):

wherein x2 = 0.60 to 0.80, y2 = 0.35 to 0.05, z2 = 0.05 to 0.15, and x2 + y2 + z2 = 1.00; 10) A process for preparing the polyimide of above 4, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7):

from 5 to 15 mol% of 4,4'-oxydianiline of the formula (8):

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$$H_2N$$
  $O$   $NH_2$  (8)

and from 30 to 10 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units represented by the formula (13):

wherein x3 = 0.65 to 0.75, y3 = 0.30 to 0.10, z3 = 0.05 to 0.15, and x3 + y3 + z3 = 1.00;

11) A process for preparing the polyimide of above 5, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7):

$$H_2N \longrightarrow NH_2$$
 (7)

from 30 to 10 mol% of metaphenylenediamine of the formula (10):

and from 5 to 15 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

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in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units represented by the formula (14):

wherein x4 = 0.65 to 0.75, y4 = 0.05 to 0.15, z4 = 0.30 to 0.10, and x4 + y4 + z4 = 1.00;

12) A polyamic acid varnish, which contains a precursor of the polyimide of above 1, or that is, a random copolymer, polyamic acid having repeating units represented by the formula (15):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00;

- 13) A polyimide film containing the polyimide of above 1;
- 14) A polyimide circuit substrate, which is a laminate of the polyimide of above 1 and metal leaf.

# BEST MODES OF CARRYING OUT THE INVENTION

[0013] The polyimide of the invention is a polyimide copolymer of formula (1), precisely, formulae (2) to (5).

[0014] Indicating the polyimide copolymer, these formulae show one repeating unit of the copolymer formed through reaction of pyromellitic acid dianhydride with the respective diamines, and the repeating unit shows the composition ratio of the polyimide units that constitute the copolymer. Therefore, in the copolymer of the invention, the constitutive polyimide units are not specifically defined in point of the sequence regularity and the sequence order, but are at random.

[0015] In the invention, the tetracarboxylic dianhydride component essential to the polyimide copolymer is pyromellitic acid dianhydride of the formula (6):

[0016] The acid dianhydride is selected and combined with a diamine component mentioned below to give a polyimide copolymer, and the resulting polyimide has the desired linear expansion coefficient.

[0017] Any other tetracarboxylic dianhydride may be used herein, not interfering with the effect of the invention, but its amount is generally at most 10 mol% of all the tetracarboxylic dianhydride used herein, preferably at most 5 mol%

[0018] Concrete examples of the additional tetracarboxylic dianhydrides are, for example,

15 3,3',4,4'-biphenyltetracarboxylic dianhydride,

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2,3,3',4'-biphenyltetracarboxylic dianhydride,

3,3',4,4'-benzophenonetetracarboxylic dianhydride,

bis(3,4-dicarboxyphenyl)ether dianhydride,

bis(3,4-dicarboxyphenyl)sulfide dianhydride,

bis(3,4-dicarboxyphenyl)sulfone dianhydride,

bis(3,4-dicarboxyphenyl)methane dianhydride,

2,2-bis(3,4-dicarboxyphenyl)propane dianhydride,

2,2-bis(3,4-dicarboxyphenyl)-1,1,1,3,3,3-hexafluoropro pane dianhydride,

1,3-bis(3,4-dicarboxyphenoxy)benzene dianhydride,

1,4-bis(3,4-dicarboxyphenoxy)benzene dianhydride,

4,4'-bis(3,4-dicarbophenoxy)biphenyl dianhydride,

2,2-bis[(3,4-dicarboxyphenoxy)phenyl]propane dianhydride,

2,3,6,7-naphthalenetetracarboxylic dianhydride,

1,4,5,8-naphthalenetetracarboxylic dianhydride.

butane-1,2,3,4-tetracarboxylic dianhydride,

pentane-1,2,4,5-tetracarboxylic dianhydride,

cyclobutanetetracarboxylic dianhydride,

cyclopentane-1,2,3,4-tetracarboxylic dianhydride,

cyclohexane-1,2,4,5-tetracarboxylic dianhydride,

cyclohexa-1-ene-2,3,5,6-tetracarboxylic dianhydride,

3-ethylcyclohexa-1-ene-3-(1,2),5,6-tetracarboxylic dianhydride,

1-methyl-3-ethylcyclohexane-3-(1,2),5,6-tetracarboxyli c dianhydride,

1-methyl-3-ethylcyclohexa-1-ene-3-(1,2),5,6-tetracarbo xylic dianhydride,

1-ehtylcyclohexane-1-(1,2)3,4-tetracarboxylic dianhydride,

1-propylcyclohexane-1-(2,3),3,4-tetracarboxylic dianhydride,

1,3-dipropylcyclohexane-1-(2,3),3-(2,3)-tetracarboxyli c dianhydride,

dicyclohexyl-3,4,3',4'-tetracaroxylic dianhydride, bicyclo[2.2.1]heptane-2,3,5,6-tetracarboxylic dianhydride,

bicyclo[2.2.2]octane-2,3,5,6-tetracarboxylic dianhydride,

45 bicyclo[2.2.2]octo-7-ene-2,3,5,6-tetracarboxylic dianhydride, and the like.

[0019] These tetracarboxylic acid dianhydrides may be used herein either singly or as combined at the same time. [0020] On the other hand, the diamine component to form the polyimide copolymer of the invention is as follows:

a) The essential diamine is paraphenylenediamine (also referred to as 1,4-diaminobenzene) of the formula (7):

$$H_2N - \bigcirc -NH_2$$
 (7)

b) The other diamines to be combined with the essential paraphenylenediamine are:

4,4'-oxydianiline (also referred to as 4,4'-diaminodiphenyl ether) of the formula (8):

$$H_2N - O - O - NH_2$$
 (8)

metaphenylenediamine (also referred to as 1,3-diaminobenzene) of the formula (10):

and diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

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H<sub>2</sub>N NH<sub>2</sub> (12)

20 [0021] Of these diamines, the diamine of formula (8) alone, or two diamines of formulae (8) and (10), or formulae (10) and (12) or formulae (8) and (12) are combined with the diamine of the formula (7).

[0022] The diaminomethyl-bicyclo[2.2.1]heptanes may be in any form of their isomers, but are essentially 2,5-diaminomethyl-bicyclo[2.2.1]heptane or 2,6-diaminomethyl-bicyclo[2.2.1]heptane, including their stereoisomers.

[0023] The diaminomethyl-bicyclo[2.2.1]heptanes may be in the form of a mixture of their isomers.

[0024] The polyimide copolymer of the invention is obtained through thermal imidation of a random copolymer, polyamic acid copolymer prepared by reacting the tetracarboxylic acid dianhydride mentioned above with two or three diamines also mentioned above, and it is represented by the formula (1), precisely formulae (2), (3), (4) and (5).

[0025] Specifically, the polyimide copolymer of the invention includes the following:

<1> A polyimide copolymer having repeating units represented by the formula (1):

wherein R<sub>1</sub>, R<sub>2</sub>, x, y and Z have the same meanings as above, which is obtained by reacting 1 equivalent mol of a tetracarboxylic acid dianhydride, pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of phenylenediamine of the formula (7):

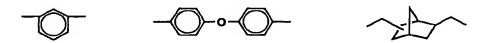
$$H_2N \longrightarrow NH_2$$
 (7)

and from 40 to 20 mol% of diamines of the formulae (16) and (17):

$$H_2N - R_1 - NH_2 \tag{16}$$

$$H_2N - R_2 - NH_2$$
 (17)

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from



and R<sub>1</sub> and R<sub>2</sub> may be the same or different,

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in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer having repeating units represented by the formula (15):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00; <2> A polyimide copolymer having repeating units represented by the formula (2):

$$\frac{1}{\left( \begin{array}{c} 0 \\ 0 \\ 0 \end{array} \right)} = \left( \begin{array}{c} 0 \\ 0 \\ 0 \end{array} \right) = \left( \begin{array}{c} 0 \\$$

wherein x1 and y1 have the same meanings as above, which is obtained by reacting 1 equivalent mol of a tetracarboxylic acid dianhydride, pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of the formula (7):

$$H_2N \longrightarrow NH_2$$
 (7)

and from 0.20 to 0.40 mol% of 4,4'-oxydianiline of the formula (8):

$$H_2N - O - O - NH_2$$
 (8)

followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer having repeating units represented by the formula (9):

wherein x1 and y1 have the same meanings as above;

<3> A polyimide copolymer having repeating units represented by the formula (3):

$$\left( \bigcirc \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{XZ} \left( \bigcirc \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{YZ} \left( \bigcirc \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{YZ} \left( \bigcirc \stackrel{\circ}{N} \stackrel{\circ}{C} \stackrel{\circ}{N} \right)_{ZZ} (3)$$

wherein x2, y2 and z2 have the same meanings as above, which is obtained by reacting 1 equivalent mol of a tetracarboxylic acid dianhydride, pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of the formula (7), from 5 to 15 mol% of 4,4'-oxydianiline of the formula (8), and from 5 to 35 mol% of metaphenylenediamine of the formula (10):

followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer represented by the formula (11):

preferably a polyimide copolymer of formula (3) wherein x2 = 0.60 to 0.80, y2 = 0.30 to 0.10, z2 = 0.10, and x2 + y2 + z2 = 1.00, which is obtained by reacting 1 equivalent mol of pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7), 10 mol% of 4,4'-oxydianiline of the formula (8) and from 25 to 15 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12), in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer;

<4> A polyimide copolymer having repeating units represented by the formula (4):

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wherein x3, y3 and z3 have the same meanings as above, which is obtained by reacting 1 equivalent mol of a tetracarboxylic dianhydride, pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7), from 10 to 30 mol% of 4,4'-oxy-dianiline of the formula (8), and from 25 to 15 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer represented by the formula (13):

preferably a polyimide copolymer of the formula (4) wherein x3 = 0.65 to 0.75, y3 = 0.25 to 0.15, z3 = 0.10, and x3 + y3 + z3 = 1.00, which is obtained by reacting 1 equivalent mol of pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7), from 15 to 25 mol% of metaphenylenediamine of the formula (10) and 10 mol% of diaminomethylbicyclo[2.2.1]heptanes of the formula (12), in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer;

<5> A polyimide copolymer having repeating units represented by the formula (5):

wherein x4, y4 and z4 have the same meanings as above, which is obtained by reacting 1 equivalent mol of a tetracarboxylic acid dianhydride, pyromellitic acid dianhydride with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7), from 25 to 15 mol% of metaphenylenediamine of the formula (10) and 10 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12), followed by thermally imidizing the resulting random copolymer, polyamic acid copolymer represented by the formula (14):

preferably a polyimide copolymer of the formula (5) wherein x4 = 0.65 to 0.75, y4 = 0.10, z4 = 0.25 to 0.15, and x4 + y4 + z4 = 1.00, which is obtained by reacting 1 equivalent mol of pyromellitic acid dianhydride with from 0.9 to 1.1

equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine, from 25 to 15 mol% of metaphenylenediamine and 10 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formulae (7), (10) and (12), respectively, in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid

[0026] The polyimide copolymer of the invention has repeating units mentioned above, and its linear expansion coefficient at 100 to 200°C is in the range of 10 to 20 ppm/K.

[0027] In general, the molecular weight of the polyimide copolymer of the invention is controlled by controlling the blend ratio of the starting tetracarboxylic acid dianhydride and diamine compounds for the copolymer. Generally, the molar ratio of all the diamine component to all the acid dianhydride component in producing the polyimide copolymer must be made in the range of 0.9 to 1.1.

[0028] The polymer molecule terminals of the polyimide copolymer represented by any of the above-mentioned formulae may be blocked or not.

[0029] Preferably, the polymer molecule terminals are blocked with a group not reactive with amines or dicarboxylic anhydrides. Concretely, it is desirable that the polymer molecule terminals of the polyamic acid copolymer and the polyimide copolymer having repeating units of formulae mentioned above are blocked with an aromatic dicarboxylic anhydride represented by the formula (16):



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wherein  $Z_1$  is a divalent group having from 6 to 15 carbon atoms, and selected from the group consisting of a monocyclic aromatic group, a condensed polycyclic aromatic group, and a non-condensed polycyclic aromatic group which is composed of aromatic groups mutually bonded to each other either directly or via a crosslinking member,

[0030] Concretely, the aromatic dicarboxylic anhydride of the formula (16) includes, for example,

phthalic anhydride,

4-phenylphthalic anhydride,

4-phenoxyphthalic anhydride,

4-phenysulfinylphthalic anhydride,

4-phenylsulfonylphthalic anhydride,

4-phenylcarbonylphthalic anhydride,

4-(2-phenylisopropyl)phthalic anhydride,

4-(1,1,1,3,3,3-hexafluoro-2-phenylisopropyl)phthalic anhydride, 1,8-naphtalenedicarboxylic anhydride and the

[0031] These aromatic dicarboxylic anhydrides are used herein either singly or as combined.

[0032] The amount of the aromatic dicarboxylic anhydride to be used is in the range of 0.001 to 1.0 mol per mol of all the diamine compound. Preferably, it is in the range 0.01 to 0.5 mols.

[0033] The aromatic monoamine of the formula (17) includes, for example, aniline, toluidines, xylidines, chloro-anilines, bromo-anilines, nitro-anilines, amino-phenols, anisidines, phenetidines, aminobenzaldehydes, aminobenzonitriles, aminobiphenyls, aminophenylphenylethers, aminobenzophenones, aminophenylphynolsulfides, aminophenylphenylsulfones, naphthylamines, aminonaphthols and aminoanthracenes,

[0034] These aromatic monoamines may be substituted with a substituent not reactive with amines and dicarboxylic acid anhydrides. These aromatic monoamines may be used herein either singly or as combined.

[0035] Like that of the aromatic dicarboxylic acid anhydride, the amount of the aromatic monoamine to be used is in the range of 0.001 to 1.0 mol per mol of all the tetracarboxylic dianhydride. Preferably, it is in the range of 0.01 to 0.5 mols.

[0036] The mode of blocking the molecular terminals of the polyimide copolymer of the invention is as follows:

55 <1> In one case, an excess of the diamine compound is used and the terminals are blocked with an aromatic dicarboxylic anhydride. In this case, the amount of the tetracarboxylic dianhydride is from 0.9 to smaller than 1.0 mol and that of the aromatic dicarboxylic anhydride is from 0.001 to 1.0 mol, per mol of the diamine compound. <2> In the other case, an excess of the tetracarboxylic dianhydride is used and the terminals are blocked with an

aromatic monoamine. In this case, the amount of the diamine compound is from 0.9 to smaller than 1.0 mol and that of the aromatic monoamine is from 0.001 to 1.0 mol, per mol of the tetracarboxylic acid dianhydride.

[0037] In case where the molar ratio of all the diamine compound to the tetracarboxylic dianhydride is defined to be in the range of 0.9 to 1.1, the molecular weight of the polyamic acid to be obtained will be as follows:

Specifically, the inherent viscosity of the polyamic acid, measured in a solvent of N-methyl-2-pyrrolidone having the acid concentration of 0.5 g/dl at 35°C, is in the range of 0.1 to 3.0 dl/g.

[0038] The polyimide of the invention is a copolymer, and the configuration of two or more repeating units that constitute is not specifically defined in point of the sequential control and the regularity of the repeating units, or that is, the copolymer is a random copolymer. Accordingly, when diamines and tetracarboxylic dianhydrides of three or more different types in all are reacted to produce the copolymer, the order of adding the monomers to a reactor is not specifically defined, and the monomers may be added thereto all at a time or successively in divided portions.

[0039] The reaction to form the polyamic acid copolymer is generally carried out in an organic solvent. The organic solvent may be any ordinary one, including

- a) phenol based solvent; phenol, o-chlorophenol, m-chlorophenol, p-chlorophenol, o-cresol, m-cresol, p-cresol, 2,3-xylenol, 2,4-xylenol, 2,5-xylenol, 3,5-xylenol, 3,5-xylenol, and the like,
- b) aprotic amide based solvent; N, N-dimethylformamide, N, N-dimethylacetoamide, N, N-diethylacetoamide, N-methyl-2-pyrolidone, 1, 3-dimethyl-2-imidazolodinone, N-methyl caprolactam, hexamethylphosphotriamide and the like

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- c) ether based solvent; 1, 2-dimethoxyethane, bis(2-methoxyethyl)ether, 1,2-bis(2-methoxyethoxy)ethane, tetrahydrofuran, bis[2-(2-methoxyethoxy)ethyl]ether, 1,4-dioxane and the like,
- d) amine based solvent; Pyridine, quinoline, isoquinoline,  $\alpha$ -picoline,  $\beta$ -picoline,  $\gamma$ -picoline, isophorone, piperidine, 2,4-rutidine, 2,6-rutidine, trimethylamine, triethylamine, tripropylamine, tributylamine and the like, and
- e) other solvent; dimethylsulfoxide, dimethylsulfone, diphenylether, sulforan, diphenylsulfone, tetramethylurea, anisole and the like.

[0040] These solvents may be used either singly or as combined. Of those, preferred are aprotic amide solvents; and most preferred are N,N-dimethylformamide, N,N-dimethylacetamide, and N-methyl-2-pyrrolidone.

[0041] The concentration of the reaction to be carried out in these solvents (this is hereinafter referred to as a polymerization concentration) is not specifically limited at all.

[0042] In the invention, the polymerization concentration in a solvent is defined as the ratio in terms of percentage of the overall weight of all the diamine and all the tetracarboxylic acid dianhydride used to the sum total of the overall weight of all the solvent used and the overall weight of all the diamine and all the tetracarboxylic dianhydride used. Preferably, the polymerization concentration is in the range of 5 to 40 %, more preferably in the range of 10 to 30 %.

[0043] In the reaction to give the polyimide copolymer precursor, polyamic acid copolymer, the especially preferred reaction temperature, reaction time and reaction pressure are not specifically limited, and any known conditions may apply to these. Generally speaking, the reaction temperature preferably is in the range of -10 to 100°C or so, more preferably in the range of around the freezing temperature to 60°C or so, and most preferably in practice, it is in the range of 50 to 60°C. The reaction time varies depending on the type of the monomers used, the type of the solvent used and the reaction temperature, but preferably is in the range of 1 to 48 hours, more preferably in the range of 2 or 3 hours to over ten hours or so, and most preferably in practice, it is in the range of 4 to 10 hours. For the reaction pressure, normal pressure will be enough.

[0044] When the polyamic acid copolymer is analyzed for its molecular weight profile through gel permeation chromatography (GPC), it is desirable that the number-average molecular weight (Mn) of the copolymer is at least 45,000, more preferably is in the range of 45,000 to 200,000, the weight-average molecular weight (Mw) thereof is at least 60,000, more preferably is in the range of 60,000 to 450,000, and the molecular weight distribution thereof is in the range of 1.6 to 2.3.

[0045] The polyamic acid copolymer obtained according to the process mentioned above is dehydrated under heat (for thermal imidation) while removing the solvent from it to be a polyimide copolymer.

[0046] Concretely, a varnish of the polyamic acid copolymer is applied onto an inorganic substrate such as metal leaf or glass, using a coater or the like. The thickness of the polyamic acid copolymer varnish film to be formed on the substrate varies, depending on the solid concentration of the varnish, but is so controlled that the thickness of the polyimide film layer to be formed after the solvent removal and imidation is at most 1 mm, preferably at most 100  $\mu$ m, more preferably at most 50  $\mu$ m. The method of solvent removal is not specifically limited. For example, it is carried out under reduced pressure or in an inert atmosphere of nitrogen, helium, argon or the like, at a temperature not lower than the boiling point of the solvent used and enough for imidation. Concretely, when an aprotic amide based solvent is used, the temperature may be 200°C or higher. The baking time is not also specifically defined. In general, it may

be 2 hours or longer.

[0047] The circuit substrate of the invention is a laminate structure of the polyimide film of the invention and metal leaf, and on the side of the metal leaf of the laminate, a circuit is formed. The metal leaf for it may be any and every one generally used in flexible circuit substrates. In general, it is copper leaf or stainless steel leaf (SUS leaf). Though not specifically defined, the thickness of the metal leaf is generally at most 100 µm, preferably at most 50 µm. The surface of the metal leaf may be processed in any desired manner with no problem. For laminating the polyimide film and metal leaf, any known method is employable. Generally known are <1> a method of applying a polyamic acid varnish onto the surface of metal leaf followed by heating it for solvent removal and imidation to form a laminate of polyimide and metal leaf; and <2> a method of previously preparing a polyimide film followed by sticking it to metal leaf with a known adhesive to form a laminate of polyimide and metal leaf.

#### **EXAMPLES**

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[0048] The invention is described in more detail with reference to the following Examples, which, however, are not intended to restrict the scope of the invention. The physical properties of the products were measured according to the methods mentioned below. Inherent Viscosity (ninh):

A polyamic acid to be measured is dissolved in a solvent of N-methyl-2-pyrrolidone to have a solid concentration of 0.5 g/dl. The polymer solution is measured at 35°C.

20 E-type Mechanical Viscosity:

[0049] Measured at 25°C by using a Toki Sangyo's E-type mechanical viscometer.

Molecular Weight:

[0050] Measured at 50°C by using a Showa Denko's GPC SHODEX SYSTEM-21 with a column GPC-805. The eluent is LiBr (6 mM)-85 % phosphoric acid (6 mM) 5 DM solution; the monitor wavelength is 270 nm; and the sample flow rate is 1.0 ml/ml. Glass Transition Temperature (Tg):

Measured in nitrogen by using a Seiko's thermal analyzer, DSC 3000 Series.

5 % Weight Loss Temperature (Td5):

[0051] Measured in air by using a Seiko's thermal analyzer, DTA-TG 3000 Series.

35 Linear Expansion Coefficient:

[0052] Measured in nitrogen in a temperature range of 100 to 200°C by using a Mac Science's thermal analyzer, TMA Series.

40 Tensile Strength (TS), Tensile Modulus (TM), Elongation (EL):

[0053] Using a Shimadzu Seisakusho's tensile tester EZ-TEST (for 100 N), the sample is tested.

Dielectric Constant:

[0054] Measured at room temperature, according to ASTM-150D.

Volume Resistivity, Surface Resistivity:

50 [0055] Measured at room temperature by using a Hewlett-Packard's surface resistivity and volume resistivity meter.

Storage Stability of Varnish:

[0056] The varnishes obtained in Examples and Comparative Examples are stored in a refrigerator at 5°C for 3 months, and their condition is visually checked. With no change, the samples are good (Ο); partly solidified and whitened, the samples are not so good (Δ); and completely solidified and gelled, the samples are bad (×).

#### Example 1:

[0057] 6.01 g (0.030 mols) of 4,4'-oxydianiline (hereinafter ODA), 7.57 g (0.070 mols) of paraphenylenediamine (hereinafter pPD) and 120.68 g of a solvent, N-methyl-2-pyrrolidone (hereinafter NMP) having a water content of 300ppm were fed into a flask equipped with a stirrer, a nitrogen-introducing duct, and completely dissolved. Next, this was once cooled to around room temperature, and 21.59 g (0.099 mols) of pyromellitic acid dianhydride (hereinafter PMDA) was added thereto, while watching the temperature elevation. In this stage, the inner temperature rose from around room temperature to 55°C. The contents of the reactor were washed with 20 g of NMP, and after completely dissolved, they were reacted at 60°C for 6 hours. In this stage, the contents became a highly viscous varnish. After the reaction, the varnish was filtered under pressure through a 5  $\mu$ -filter to obtain a polyamic acid varnish. The polyamic acid had an inhrernt viscosity ( $\eta$ inh) of 1.24 dVg, an E-type mechanical viscosity of 66800 mPa-s, a number-average molecular weight (Mn) of 102600, a weight-average molecular weight (Mw) of 204100, and a molecular weight distribution (Mw/Mn) of 1.99.

[0058] The polyamic acid varnish was cast on a glass sheet to form a layer thereon having a thickness of 0.20 mm, heated in a nitrogen-purged inert oven from room temperature up to 250°C over a period of 2 hours, and baked at 250°C for 2 hours. After cooled to room temperature, the polyimide film formed was peeled from the glass sheet, and its thickness was measured and was 30 µm. In its thermal analysis, the polyimide film had no glass transition temperature (Tg) and its 5 % weight loss temperature (Td5) was 534°C in air. The linear expansion coefficient of the polyimide film was 10 ppm/K in a temperature range of 100 to 200°C.

[0059] Regarding its mechanical properties, the polyimide film had a tensile strength (TS) of 16.4 kgf/mm², a tensile modulus (TM) of 396 kgf/mm², and an elongation (EL) of 11 %. Regarding its electric properties, the polyimide film had a dielectric constant of 3.7, a volume resistivity of larger than 10<sup>16</sup> Ω·cm, and a surface resistivity of larger than 10<sup>16</sup> Ω. [0060] After stored in a refrigerator at 5°C for 3 months, the polyamic acid varnish obtained in this Example did not change at all.

[0061] The test data of this Example are given in Table 1 along with the test data of the other Examples.

#### Examples 2 to 12:

[0062] Polyamic acid varnishes and polyimide films were obtained in the same manner as in Example 1, for which, however, the compositions of the diamine components to be used were varied as in Table 1. The test data of these Examples are summarized in Table 1.

#### Comparative Example 1:

[0063] A polyamic acid varnish was obtained in the same manner as in Example 1, for which, however, the diamine component to be used was varied to pPD, 10.81 g (0.100 mols). The polyamic acid had ηinh of 0.88 dl/g, and an E-type mechanical viscosity of 38500 mPa-s.

[0064] Forming a film of the polyamic acid varnish was tried in the same manner as in Example 1, but a polyimide film enough for evaluation could not be obtained since the varnish did not have film-forming ability. The test data of Comparative Example 1 are summarized in Table 2.

# Comparative Examples 2 to 18:

[0065] Polyamic acid varnishes and polyimide films (some varnishes did not have film-forming ability) were obtained in the same manner as in Example 1 and Comparative Example 1, for which, however, the compositions of the diamine components in Comparative Example 1 to be used were varied as in Table 2. The test data of these Comparative Examples are summarized in Table 2.

#### Example 13:

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[0066] The polyamic acid varnish obtained in Example 1 was cast on copper leaf of 25 µm thick (its linear expansion coefficient at 100 to 200°C is 17 ppm/K) to form a layer thereon having a thickness of 0.20 mm, heated in a nitrogen-purged inert oven from room temperature up to 250°C over a period of 2 hours, and baked at 250°C for 2 hours. After cooled to room temperature, obtained was a laminate of polyimide film and copper leaf. Not warped, deformed and curled, the laminate was almost flat. The test data of this Example are given in Table 3.

# Examples 14 to 16:

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[0067] The polyamic acid varnishes obtained in Examples 5, 7 and 10 were separately cast on copper leaf and processed in the same manner as in Example 13 to obtain laminates of polyimide film and copper leaf. Like that of Example 13, all these laminates were almost flat, not warped, deformed and curled. Like those of Example 13, the test data of these Examples are given in Table 3.

Comparative Examples 19 to 25:

[0068] The polyamic acid varnishes obtained in Comparative Examples 3 to 5, 11, 14, 16 and 18 were separately cast on copper leaf in the same manner as in Example 13 to obtain laminates of polyimide film and copper leaf. The laminates were warped, deformed and curled, and were not flat. Along with those of Examples 13 to 16, the test data of these Comparative Examples are given in Table 3.

5	rs (kgf/mm²)	16.4	16.0	18.2	15.0	15.6	9.7	
10	LINEAR EXPANSION COEFFICIENCY (PDM/K)	10	12	10	14	15	13	
	7d5 (C)	534	534	536	530	530	530	Ī
20	<sup>Tg</sup> (Ĉ)	QN.	S	ON.	8	S	QN	
30	E-TYPE MECHANICAL VISCOSITY (mPa/s)	66800	44800	55000	15300	20500	26500	
	η inh (d1/g)	1.24	1.12	1.18	0.75	0.80	0.84	
35								
40	POLYMER CONCENTRA (wt.8)	18	<b>←</b>	<b>←</b>	<b>←</b>	<b>←</b>	<b>←</b>	
45 50	E DIAMINE COMPOSITION CONCENTRATION (WT%)	pPD/ODA=0.70/0.30	1 =0.60/0.40	1 =0.80/0.20	PPD/mPD/ODA =0.60/0.30/0.10	f =0.70/0.20/0.10	† =0.80/0.10/0.10	ADD/Nana /ODa
g K E	PLE	1	2	3	4	2	9	

								-
XAMPLE 10.	DIAMINE COMPOSITION	POLYMER CONCENTRATION (WL8)	η inh (d1/g)	E-TYPE MECHANICAL VISCOSITY (mPa/s)	<sub>2</sub>	7d5 (2)	LINEAR EXPANSION COEFFICIENCY (DCM/K)	TS (kgf/mm²)
1	pPD/ODA=0.70/0.30	18	1.24	66800	2	534	10	16.4
2	1 =0.60/0.40	<b>←</b>	1.12	44800	Q	534	12	16.0
3	1 =0.80/0.20	<b>←</b>	1.18	55000	Q.	536	10	18.2
4	PPD/mPD/ODA =0.60/0.30/0.10		0.75	15300	<u>8</u>	530	14	15.0
5	† =0.70/0.20/0.10	Ţ	08.0	20500	8	530	15	15.6
9	+ =0.80/0.10/0.10	<b>↓</b>	0.84	26500	QN	530	13	9.7
7	pPD/NBDA/ODA =0.70/0.20/0.10	20	0.68	10650	QN	477	16	11.4
80	f =0.65/0.25/0.10	₽.	0.74	14700	Q	468	18	12.3
6	† =0.75/0.15/0.10	<b>+</b>	0.82	16300	Ð	507	14	13.3
10	pPD/mPD/NBDA =0.70/0.20/0.10	<b>←</b>	0.65	9700	Q	488	18	12.1
11	f =0.65/0.25/0.10	<b>L</b>	0.61	8800	£	486	19	12.3
12	† =0.75/0.15/0.10	<b>.</b>	0.70	10800	ND	502	16	13.2

	Description of Terms:
5	ηinh: inherent viscosity
	Tg: glass transition temperature
10	Td5: 5 % weight loss temperature
	TS: tensile strength
15	pPD: paraphenylenediamine
	ODA: 4,4'-oxydianiline
20	mPD: metaphenylenediamine
	NBDA: diaminomethyl-bicyclo[2.2.1]heptanes
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35	
40	
45	
50	

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5	MW/Mn	1.99	2.00	1.90	2.00	1.90	1.85	2.12	2.01	1.95	1.80	1.72	1.90
10	Μw	204100	180000	192100	105400	115500	123500	91300	103400	119500	85200	77200	95300
15	Mn	102600	90000	101100	52700	00809	66800	43100	51400	61300	47300	44900	50200
.5	- <u>2</u> 6								-	 		-	
20	STORAGE STABILITY VARNISH	0	0	0	0	0	0	0	0	0	0	0	0
25	SURFACE RESISTIVITY (Q)	>10 <sup>16</sup>	>1016	>10 <sup>17</sup>	>1016	>1016	>1016	>10 <sup>17</sup>	>1016	>10 <sup>17</sup>	>1017	>1016	>1017
30	VOLUME RESISTIVITY (Q · cm)	>1016	>10¹²	>1017	>1017	>1016	>1016	>1016	>1016	>1017	>1017	>1016	>1016
40	DIELECTRIC COEFFICIENT	3.7	3.6	3.7	3,6	3.6	3.7	3.6	3.5	3.6	3.6	3.5	3.6
45 (pg	EL (%)	ជ	18	11	2	80	4	11	16	13	6	10	10
S & & & & & & & & & & & & & & & & & & &	TM (kgf/mm²)	396	390	463	398	426	412	231	286	298	282	294	306
55	EXAMPLE NO.	1	2	3	4	ស	9	7	8	6	10	11	12

	Desc	Sriptic	on or a	Leli	15:		
5	TM:	tensil	le modu	ulus	:		
	EL:	elonga	ation				
10	Vari	nish st	orage	sta	bilit	λ:	
	0:	No cha	nge.				
15	Δ:	Partly	soliđ	ifi	ed or	whit	ened.
	0:	Comple	tely s	011	difie	d or	gelled
20							weight
	Mw:	weight	-avera	ige	molec	ular	weight
25							
30							
35							
40							
45							
50							

			-																
5	rs (kgf/mm²)	form film	10.1	14.2	10.1	14.4		orm film		12.4	9.7	9.2		orm film	10.0	6.6	10.6	form film	10.2
	LINEAR EXPANSION COEFFICIENCY (ppm/K)	sible to	27	38	09	24		impossible to form film		33	34	49		impossible to form film	27	47	40	impossible to f	28
15	7d5 (උ)	510	517	548	436	546	540	519	523	533	517	398	418	208	443	427	418	496	488
20	7g (Ĉ)	S	Ş	N D	292	Q.	2	Ω <sub>N</sub>	S	S	S	334	S	S	QN	Ω	QN	Q.	Q.
25	E-TYPE MECHANICAL VISCOSITY (MPA·S)	38500	6930	5530	640	19100	18700	9200	7420	61900	33900	2660	16000	20200	7100	7600	115000	11500	6200
30	η inh (dl/g)	0.88	0.71	0.67	0.37	1.26	1.05	0.81	99.0	1.09	86.0	0.62	0.85	0.80	09.0	0.73	2.03	0.73	0.58
35	POLYMER CONCENTRA TION (WT%)	20	ţ	Ţ	Į	ţ	ļ	20	ţ	<b>+</b>	1	1	<b>+</b>	ļ	Į	ļ	<b>+</b>	1	1
40	NE COMPOSITION					50/0.50	. 01.0/	50/0.50	/0.20	06.0/07	/0.50	.30/0.70	/0.50	oa 70.10	0.60/0.30/0.10	.30/0.70	/0.50	DA 70.10	0.60/0.30/0.10
45	DIAMI	pPD=1.00	mPD=1.00	ODA=1.00	NBDA=1.00	pPD/ODA=0.50/0.50	1 =0.90/0.10	pPD/mPD=0.50/0.50	1 =0.80/0.20	mPD/ODA=0.70/0.30	1 =0.50/0.50	PPD/NBDA=0.30/0.70	1 =0.50/0.50	PPD/NBDA/ODA =0.80/0.10/0.10	1 0.60/0	mPD/NBDA=0.30/0.70	1 =0.50/0.50	pPD/mPD/NBDA =0.80/0.10/0.10	1 0.60/0
	COMPARATIVE	1	2	3	4	5	9	7	80	6	10	11	12	13	14	15	16	17	18
55	ОШ						,		لب										]

			-		-			-	-		-	-		-				1	
5	STORAGE STABILITY OF VARNISH	×	×	0	0	0	×	×	×	0	۵	0	۵	٥	0	0	٥	٥	0
15	SURFACE RESISTIVITY (A)	1	>1017	>1016	>1016	>1016				>1016	>1017	>1017			>1016	>1016	>10 <sup>16</sup>		>10 <sup>16</sup>
25	VOLUME RESISTIVITY (O)		>1017	>1016	>1016	>1017	orm film	יסגוו דידוויי	•	>1016	>1016	>1016	form film		>1016	>1016	>1017	form film	>1016
35	DIELECTRIC	ible to	3.8	3.6	3	3.7	impossible to form film	on arcresodim		3.6	3.6	3.4	. impossible to form film		3.4	3.6	3.5	impossible to form film	. 3.7
40	EL (%)		-	55	8	18				33	H	7			28	10	10		H
45 50	TM Tree /?	( man / = fix)	189	190	189	283				197	202	145			170	188	141		17.1
50	1 5	1	2	6	4	2	9	7	8	6	10	11	12	13	14	15	16	17	18
<i>33</i>	XX S	-		1		1	1		1	1			L	1	L	1	1	L	

b	

Example 1 Example 1 Example 1 Example 1 Used varnish parative Example 1	O O O Situation of laminate
Example 1 Example 1 Used varnish parative Example 1	O O Situation of laminate
Example 1 Used varnish parative Example 1	O Situation of laminate
Used varnish parative Example 1	Situation of laminate
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<u> </u>	×
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# INDUSTRIAL APPLICABILITY

[0069] According to the present invention, there was found out a polyimide having a suitable linear expansion coefficient that corresponds to the linear expansion coefficient of metal leaf (in this case, falling between 10 and 20 ppm/ K at a temperature falling between 100 and 200°C). More precisely, there was found out a useful polyimide which can be a polyimide circuit substrate material capable of keeping flat, neither shrinking nor expanding in its laminate.

# Claims

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1. A polyimide, which is a random copolymer having repeating units of the formula (1):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C.

2. The polyimide of claim 1, which is a random copolymer having repeating units of the formula (2):

<sup>×:</sup> Much warped, deformed and curled.

$$\left( \bigcirc \bigvee_{g} \bigvee_{x_1} \bigvee_{x_1} \left( \bigcirc \bigvee_{g} \bigvee_{x_2} \bigvee_{y_1} \right) \right) \qquad (2)$$

wherein x1 = 0.60 to 0.80, y1 = 0.40 to 0.20, and x1 + y1 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C.

3. The polyimide of claim 1, which is a random copolymer having repeating units of the formula (3):

wherein x2 = 0.60 to 0.80, y2 = 0.35 to 0.05, z2 = 0.05 to 0.15, and x2 + y2 + z2 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C.

4. The polyimide of claim 1, which is a random copolymer having repeating units of the formula (4):

wherein x3 = 0.65 to 0.75, y3 = 0.30 to 0.10, z3 = 0.05 to 0.15, and x3 + y3 + z3 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C.

5. The polyimide of claim 1, which is a random copolymer having repeating units of the formula (5):

wherein x4 = 0.65 to 0.75, y4 = 0.05 to 0.15, z4 = 0.30 to 0.10, and x4 + y4 + z4 = 1.00, and which has a linear expansion coefficient of the range of 10 to 20 ppm/K at 100 to 200°C.

- The polyimide of claim 1, of which the precursor, polyamic acid has a number-average molecular weight (Mn) of at least 40,000, a weight-average molecular weight (Mw) of at least 60,000, and a molecular weight distribution (Mw/Mn) falling between 1.6 and 2.3;
- 7. A process for preparing the polyimide of claim 1, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

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with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of a diamine of the formula (7):

and from 40 to 20 mol% of diamines of formulae (16) and (17):

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$$H_2N - R_1 - NH_2$$
 (16)

 $H_2N - R_2 - NH_2$  (17)

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and R<sub>1</sub> and R<sub>2</sub> may be the same or different,

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units of the formula (15):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from

and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00.

8. A process for preparing the polyimide of claim 2, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of a formula (7):

$$H_2N \longrightarrow NH_2$$
 (7)

and from 40 to 20 mol% of 4,4'-oxydianiline of the formula

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(8):in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units of the formula (9):

wherein x1 = 0.60 to 0.80, y1 = 0.40 to 0.20, and x1 + y = 1.00.

A process for preparing the polyimide of claim 3, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 60 to 80 mol% of paraphenylenediamine of the formula (7):

$$H_2N - \longrightarrow NH_2$$
 (7)

from 5 to 15 mol% of 4,4'-oxydianiline of the formula (8):

$$H_2N$$
  $O$   $NH_2$  (8)

and from 35 to 5 mol% of metaphenylenediamine of the formula (10):

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units of the formula (11):

wherein x2 = 0.60 to 0.80, y2 = 0.35 to 0.05, z2 = 0.05 to 0.15, and x2 + y2 + z2 = 1.00.

20 10. A process for preparing the polyimide of claim 4, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7):

from 5 to 15 mol% of 4,4'-oxydianiline of the formula (8):

and from 30 to 10 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units of the formula (13):

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wherein x3 = 0.65 to 0.75, y3 = 0.30 to 0.10, z3 = 0.05 to 0.15, and x3 + y3 + z3 = 1.00.

10 11. A process for preparing the polyimide of claim 5, which comprises reacting 1 equivalent mol of pyromellitic acid dianhydride of the formula (6):

with from 0.9 to 1.1 equivalent mols of a diamine mixture that comprises from 65 to 75 mol% of paraphenylenediamine of the formula (7):

$$H_2N \longrightarrow NH_2$$
 (7)

from 30 to 10 mol% of metaphenylenediamine of the formula (10):

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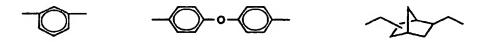
and from 5 to 15 mol% of diaminomethyl-bicyclo[2.2.1]heptanes of the formula (12):

in an organic solvent, followed by thermally imidizing the resulting random copolymer, polyamic acid having repeating units of the formula (14):

wherein x4 = 0.65 to 0.75, y4 = 0.05 to 0.15, z4 = 0.30 to 0.10, and x4 + y4 + z4 = 1.00.

12. A polyamic acid varnish, which contains a precursor of the polyimide of claim 1, or that is, a random copolymer, polyamic acid having repeating units of the formula (15):

wherein R<sub>1</sub> and R<sub>2</sub> each represent a divalent group selected from



and  $R_1$  and  $R_2$  may be the same or different; and x = 0.60 to 0.80, y + z = 0.40 to 0.20, and x + y + z = 1.00.

13. A polyimide film containing the polyimide of claim 1.

20 14. A polyimide circuit substrate, which is a laminate of the polyimide of claim 1 and metal leaf.

# INTERNATIONAL SEARCH REPORT

International application No.

PCT/JP01/06919

A. CLASS Int.	EFICATION OF SUBJECT MATTER Cl <sup>7</sup> C08G73/10, C08J5/18, C09D	179/08, H05K1/03 // C08L7	79:08					
According to	o International Patent Classification (IPC) or to both m	ational classification and IPC						
B. FIELD	SSEARCHED							
Minimum d Int.	commentation searched (classification system followed Cl <sup>7</sup> C08G73/10, C08J5/18, C09D:							
	ion searched other than minimum documentation to the							
	sta base consulted during the international search (nan ONLINE	ie of data base and, where practicable, sex	rch terms used)					
C. DOCU	MENTS CONSIDERED TO BE RELEVANT							
Category	Citation of document, with indication, where ap	ppropriate, of the relevant passages	Relevant to claim No.					
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x	JP 63-175026 A (Kanegafuchi Ch 19 July, 1988 (19.07.88), Claims; page 2, upper right colum		1,2,6-8 12-14					
Y A	Y left column, line 10; page 2, lower right column, lines 3,9 A 3, 13; working example 3 (Family: none) 4,5,10,11							
Y Y	JP 63-199236 A (Asahi Chemical 17 August, 1988 (17.08.88), Claims; page 1, lower right col lower right column, line 17; sy (Family: none)	lumn, line 1; page 3,	1,6,7,12-14 3,9					
A	JP 10-7906 A (Chisso Corporation 13 January, 1998 (13.01.98), Claims (Family: none)	on), .	1,4-7,10-14					
Further	documents are listed in the continuation of Box C.	See patent family annex.						
"A" docume consider	categories of cited documents: at defining the general state of the art which is not red to be of particular relevance	"I" later document published after the inter priority date and not in conflict with the understand the principle or theory under	e application but cited to					
"L" docume	ocument but published on or after the international filing at which may throw doubts on priority claim(s) or which is establish the publication date of another citation or other	document of particular relevance; the c considered novel or cannot be considered step when the document is taken alone occument of particular relevance; the c	laimed invention cannot be ed to involve an inventive					
special:	catalitist the publication date of another clistical or other reason (as specified) at referring to an oral disclosure, use, exhibition or other	considered to involve an inventive step combined with one or more other such	when the document is documents, such					
"P" docume	nt published prior to the international filing date but later priority date claimed	"&" document member of the same patent if						
	ctual completion of the international search ugust, 2001 (29.08.01)	Date of mailing of the international scarce 11 September, 2001 (						
	ailing address of the ISA/ nese Patent Office	Authorized officer						
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# INTERNATIONAL SEARCH REPORT

International application No.
PCT/JP01/06919

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C (Continua	tion). DOCUMENTS CONSIDERED TO BE RELEVANT		
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A	US 4874835 A (General Electric Company), 17 October, 1989 (17.10.89), Formula (II), (III), (IV), & JP 1-131245 A & GB 2222409 A & FR 2620453 A & DE 3830064 A & US 4855391 A	herzań Kes	3,9
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